

average Mn–O bond length of $1\cdot61 \pm 0\cdot01$ Å. The e.s.d. is here set at a somewhat higher value to allow for the effect of some possible systematic errors. This bond length is in close agreement with the result of 1·629 Å found in KMnO_4 by Palenik after applying a correction of 0·022 Å for anisotropic motion. Such a correction cannot be made in the present study on account of the incomplete thermal motion data.

The structure as seen in the [001] projection seems to consist of stacks of MnO_4 tetrahedra and rows of Ag atoms extending in the c direction. The difference in z coordinates of the Ag and Mn atoms closest in projection is almost exactly $\frac{1}{2}$, and leads to a regular zigzag Ag–Mn–Ag–Mn chain running in the c direction. This description probably has no significance since there are closer Ag–Mn (and Ag–O) approaches in other directions. Each Ag atom is in fact surrounded by seven MnO_4 groups with Ag–Mn distances varying between 3·51 and 3·81 Å, and corresponding close Ag–O approaches between 2·33 and 2·67 Å. The shortest Ag–Ag and Mn–Mn distances are 3·63 and 4·33 Å respectively, while the closest approaches between oxygen atoms in different permanganate groups vary from 2·81 to 3·06 Å. The shortest Ag–O and non-bonded O–O distances are markedly shorter than the corresponding distances in KMnO_4 (2·62 and 3·06 Å respectively). The present structure is clearly more compact since the molecular volume of AgMnO_4 is about 15% smaller than that of KMnO_4 . The closer approaches found in AgMnO_4 compared to KMnO_4 can probably be attributed to the fact that AgMnO_4 is not a truly ionic compound, but contains a measure of covalent bonding. Unfortunately, accurate data for similar permanganates are not available.

The investigation of the structural aspects of the thermal decomposition of silver permanganate, which led to this structure determination, was undertaken at the suggestion, and under the initial guidance of Dr F. H. Herbstein. In the present work, helpful discussions with Dr G. Gafner are gratefully acknowledged.

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Acta Cryst. (1968), B24, 1062

The Crystal Structure of Cd_3As_2

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(Received 20 September 1967)

A Weissenberg study of the structure of Cd_3As_2 has shown that the unit cell is tetragonal ($a=12\cdot67 \pm 0\cdot01$ Å, $c=25\cdot48 \pm 0\cdot02$ Å, $Z=32$, space group $I4_1cd$). Arsenic ions are approximately cubic close-packed and Cd ions are tetrahedrally coordinated. The main difference between the present structure and that proposed by von Stackelberg & Paulus lies in the disposition of the vacant tetrahedral sites. Both structures are geometrically related to the fluorite structure.

Introduction

The crystal structure of Cd_3As_2 was determined by von Stackelberg & Paulus (1935) as tetragonal ($a=8\cdot95$, $c=12\cdot65$ Å, space group $P4_2/nmc$) with As ions approximately in a cubic close-packed array. Cd ions

were tetrahedrally coordinated and each As ion was surrounded by Cd ions at six of the eight corners of a distorted cube, the two vacant sites being at diagonally opposite corners of a cube face.

In the same paper von Stackelberg & Paulus also reported the crystal structure of Zn_3As_2 as being the

same as that of Cd_3As_2 , but with somewhat smaller cell parameters. However, a single-crystal study of Zn_3As_2 by Cole, Chambers & Dunn (1956), using rotation and precession techniques, revealed a unit cell which was body-centred tetragonal, the probable space group being $I4_1/acd$ with the a and c parameters, respectively, $1/2$ and 2 times those found by von Stackelberg & Paulus.

Two specimens of Cd_3As_2 have been examined in the present work. The first was supplied by Dr D. W. G. Ballentyne, who suggested that the authors might care to re-examine the crystal structure of this material. The second was prepared by Mr G. L. Bucknell of this department as follows: Stoichiometric quantities of cadmium and arsenic were heated in a sealed evacuated silica tube at 675°C for 18 hours. The sample was then heated at 850°C for 2 hours prior to air quenching. The resulting polycrystalline mass was found to contain several single crystals sufficiently large for study by the Weissenberg technique. Both

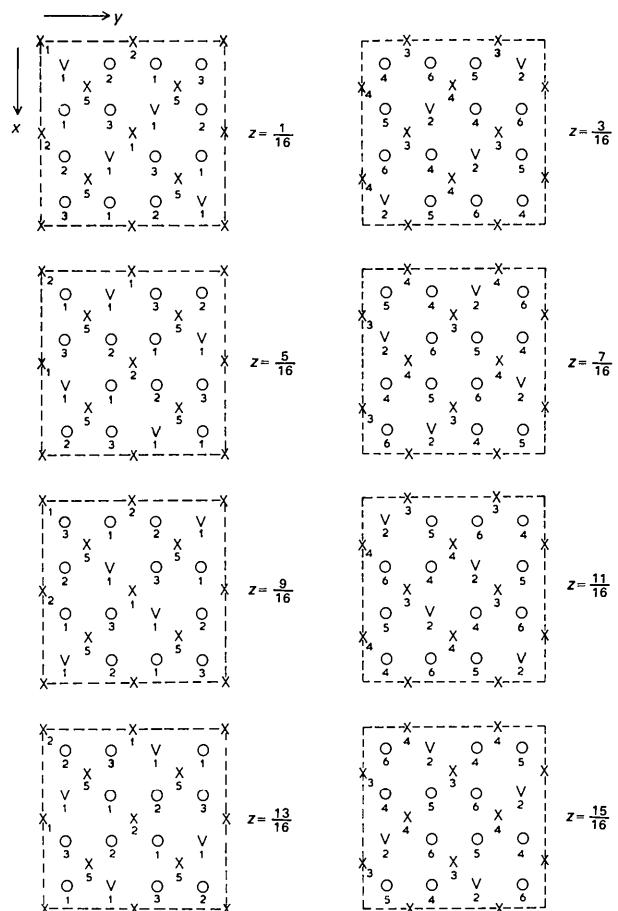


Fig. 1. Idealized structure of Cd_3As_2 . Circles represent Cd atoms in the plane of the diagram. Crosses represent As atoms at $\Delta z = \frac{1}{16}$ below the plane of the diagram. V1 and V2 are vacant Cd sites in first and second sets, respectively, of equivalent positions.

Table 1. Observed and calculated powder data

In the Table $I_c \propto |F_c|^2 p f_1(\theta)$, where p is the multiplicity and $f_1(\theta)$ is the angular term defined by Goodyear & Duffin (1957). No correction has been made for absorption since this is nearly constant for all reflexions.

For the observed powder reflexions the reliability index

$$\Sigma |||I_o| - |I_c|| / \Sigma |I_o| = 0.090$$

d_o	d_c	hkl	I_c	ΣI_c	I_o
7.35 Å	7.33 Å	112	49	49	57
5.55	5.54	121	48	48	43
4.72	4.71	123	127	127	94
3.831	{ 3.837 3.822 }	116 132	137 225	{ 362 }	465
3.785	3.790	125	206	206	149
3.664	3.664	224	933	933	1114
3.482	3.482	231	66	66	41
3.248	3.247	233	317	317	297
3.186	3.185	008	169	{ 516 }	586
3.167	3.167	040	347	{ 516 }	586
3.063	3.063	127	44	44	36
2.914	{ 2.915 2.908 }	136 332	278 83	{ 361 }	316
2.892	{ 2.893 2.889 }	235 143	7 486	{ 493 }	414
2.527	2.528	237	70	70	40
2.442	{ 2.443 2.439 }	336 152	131 17	{ 148 }	130
2.242	{ 2.246 2.240 }	048 440	1824 910	{ 2734 }	2847
2.141	{ 2.144 2.142 }	12,11 352	9 12	{ 39 }	29
1.932	{ 1.934 1.934 }	23,11 23,11	14 14	{ 500 }	537
	1.927	1.927	5		
(vb)	1.912	{ 1.919 1.911 }	22,12 264	{ 142 }	
	1.853	1.853	12,13	49	
	1.842	{ 1.845 1.844 }	455 363	{ 4 }	161
	1.831	{ 1.842 1.832 }	448 11,14	{ 47 }	130
(vb)	1.776	{ 1.774 1.774 }	11,14 552	{ 61 }	161
		1.771	11,14	{ 11 }	115
		1.771	365	{ 42 }	
(vb)	1.707	{ 1.712 1.705 }	23,13 273	{ 75 }	200
	1.656	{ 1.657 1.651 }	13,14 556	{ 83 }	175
		1.651	68	{ 1 }	
	1.647	1.647	275	283	306
	1.593	1.593	00,16	107	
	1.583	1.584	080	323	221
			216		

specimens were found to give identical X-ray powder patterns, and the powder data given in Table 1 are based on measurements of the patterns of both samples.

Since von Stackelberg and Paulus employed a rather different method of preparation (hydrogen and nitrogen gas transfer techniques) there is a small possibility that the disagreement between their structural data and those reported here is due to a difference in crystal structure rather than in quality of X-ray technique.

Weissenberg photographs of a suitable single crystal indicated a body-centred tetragonal cell approxi-

mately four times the size of the cell found by von Stackelberg & Paulus. An accurate determination of the lattice parameters, obtained from powder data, taken with Cu K α radiation and a crystal-focusing camera of effective diameter 22.9 cm, gave $a = 12.67 \pm 0.01$ and $c = 25.48 \pm 0.02$ Å. This suggests the same relation to the data of von Stackelberg and Paulus as was found by Cole *et al.* for Zn₃As₂. Calculated and observed lattice spacings are compared in Table 1.

Intensity measurements

The crystal selected for measuring intensities was of approximately square cross-section, of side 0.003 cm, and was elongated parallel to [110]. Sixteen layer lines were recorded on equi-inclination Weissenberg photographs taken about [110] with Cu K α radiation.

Systematically absent reflexions were those for which $h+k+l=2n+1$, $2h+1=4n+1$ for hhl , $l=2n+1$ for $0kl$ and possibly $h=2n+1$ for $hk0$. These suggest I₄₁cd or I₄₁acd as the space group.

The cell and possible space groups permit more than 2000 reflexions within one octant of the sphere of reflexion for Cu K α radiation. However, about 65% of these proved too weak to be recorded. Hence the intensities of 713 recorded reflexions were estimated visually from multiple film exposures using a calibrated scale. The observed intensities were corrected for Lorentz and polarization factors and were adjusted for spot shape on the upper layer lines (Phillips, 1954). Absorption corrections were made by assuming the crystal to be cylindrical, mean $\mu r = 3.4$, and applying the correction factors given by Bond (1959).

The intensities of the powder reflexions, given in Table 1 were obtained from a microdensitometer trace.

The determination of the structure

From a consideration of the cell size and diameter, 4.44 Å (Clark, 1955), of the large As ion it follows that the As ions must be cubic close-packed, or very nearly so. For one face-centred cube of As ions, the cube side would be 6.3 Å; this is about one half of the a param-

eter (12.67 Å) and one quarter of the c parameter (25.48 Å) of the unit cell. It would appear then that the cell contains 16 such cubes, i.e. 64 As ions. With 32 formula units per cell the X-ray density is 6.31 g.cm⁻³, which is in reasonable agreement with the observed macroscopic density of 6.21 g.cm⁻³ (*Handbook of Chemistry and Physics*, 1961). With the Cd ions in tetrahedral coordination, it is possible to arrange them around each As ion as in the structure of von Stackelberg & Paulus to give a structure consistent with the symmetry of the space group I₄₁cd, the essential difference between the two structures being the orientation of the vacant cadmium sites. This distribution of Cd ions was not permitted in the other suggested space group, I₄₁acd.

The ideal parameters of the proposed structure are given in column (i) of Table 2. Initial intensity calculations, using atomic scattering factors for the neutral atoms as listed in *International Tables for X-ray*

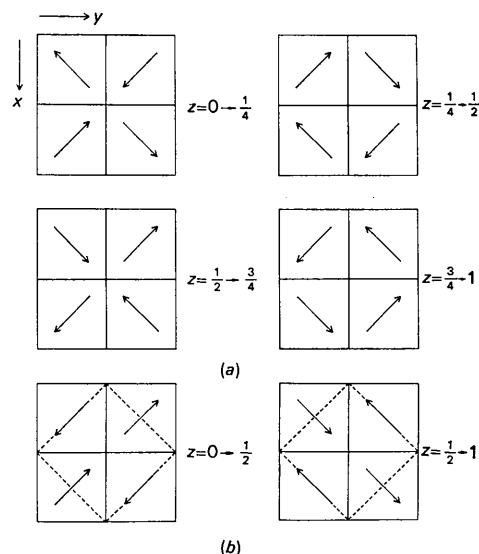


Fig. 2. Relation of the fluorite structure to (a) the present structure of Cd₃As₂ and (b) the structure proposed by von Stackelberg & Paulus. In (b) the broken line indicates the base of the unit cell.

Table 2. *Atomic parameters*

(Origin on 2). Standard deviations are given in brackets.

Atom	Equipoint	(i) Initial			(ii) Final			
		x	y	z	x	y	z	
Cd(1)	16(b)	$\frac{3}{8}$	$\frac{1}{8}$	$\frac{1}{16}$	0.3554 (11)	0.1208 (10)	0.0618 (05)	2.95 (21) Å ²
Cd(2)	16(b)	$\frac{1}{8}$	$\frac{3}{8}$	$\frac{1}{16}$	0.1188 (09)	0.3584 (09)	0.0528 (04)	2.54 (19)
Cd(3)	16(b)	$\frac{3}{8}$	$\frac{3}{8}$	$\frac{1}{16}$	0.3933 (10)	0.3933 (11)	0.0727 (04)	2.18 (14)
Cd(4)	16(b)	$\frac{1}{8}$	$\frac{1}{8}$	$\frac{3}{16}$	0.1088 (11)	0.1059 (11)	0.1772 (04)	2.35 (15)
Cd(5)	16(b)	$\frac{3}{8}$	$\frac{1}{8}$	$\frac{3}{16}$	0.3799 (10)	0.1458 (10)	0.1892 (05)	2.86 (20)
Cd(6)	16(b)	$\frac{1}{8}$	$\frac{3}{8}$	$\frac{3}{16}$	0.1417 (09)	0.3797 (08)	0.1966 (04)	2.24 (17)
As(1)	8(a)	0	0	0	0	0	0.0001 (06)	1.42 (36)
As(2)	8(a)	0	0	$\frac{1}{4}$	0	0	0.2497 (06)	1.52 (37)
As(3)	16(b)	0	$\frac{1}{2}$	$\frac{1}{8}$	0.0072 (15)	0.2540 (11)	0.1231 (10)	2.18 (35)
As(4)	16(b)	$\frac{1}{2}$	0	$\frac{1}{8}$	0.2542 (10)	-0.0055 (13)	0.1259 (10)	1.85 (32)
As(5)	16(b)	$\frac{1}{4}$	$\frac{1}{4}$	0	0.2588 (10)	0.2399 (19)	-0.0005 (12)	1.41 (21)

Crystallography (1962) and assuming an overall temperature factor of 0.47 \AA^2 , gave a reliability index, $R = \sum ||F_o| - |F_c|| / \sum |F_o|$, of about 60%. However, an inspection of the observed and calculated structure factors of the $00l$ reflexions suggested that the Cd ions

situated next to a vacant Cd site should be moved about 0.25 \AA towards the vacant site. When each Cd ion was moved 0.25 \AA towards neighbouring vacant sites, R decreased to about 30%. At this stage a three-dimensional least-squares refinement with isotropic

Table 3. Observed and calculated structure factors

The 440 and 048 single-crystal reflexions probably suffer from extinction; for these reflexions the agreement between the observed and calculated powder intensities is quite satisfactory.

H	K	L	F_o	F_c	α_c	H	K	L	F_o	F_c	α_c	H	K	L	F_o	F_c	α_c	H	K	L	F_o	F_c	α_c				
0	0	8	1838	1723	182	1	12	11	125	139	66	3	6	23	191	201	85	5	12	15	129	139	282				
0	0	16	2097	2041	0	1	12	17	23	367	0	3	6	21	172	226	273	5	13	6	288	250	38				
0	0	32	488	424	0	1	12	19	133	158	289	3	7	6	647	689	93	5	13	6	272	249	96				
0	0	64	237	211	0	1	13	2	214	233	269	3	7	6	667	655	92	5	13	10	370	349	267				
0	0	128	171	155	0	1	13	10	227	178	92	3	7	10	366	267	272	5	14	1	122	119	115				
0	0	256	1967	1768	180	1	13	18	219	202	259	3	7	14	540	638	269	5	14	5	181	216	262				
0	0	4	8	2570	4143	0	1	14	3	235	267	90	3	7	18	183	199	106	5	14	5	207	277	97			
0	0	8	465	385	0	1	14	7	99	68	225	3	7	22	300	232	94	5	15	6	292	230	62				
0	0	16	1105	1113	0	1	14	9	91	119	119	3	8	8	318	232	96	5	15	6	14	168	64				
0	0	32	176	180	0	1	15	10	240	232	262	3	8	9	235	203	103	5	16	4	967	906	0				
0	0	64	355	232	0	1	15	10	240	275	274	3	8	9	235	203	103	5	16	6	1122	1059	358				
0	0	128	130	132	0	1	15	10	240	270	274	3	8	11	162	112	233	5	16	6	357	385	357				
0	0	256	397	439	176	2	2	4	1831	1735	0	3	8	13	160	90	136	5	7	1	206	193	12				
0	0	512	331	444	359	2	2	2	13	1315	1356	3	8	15	298	207	253	5	7	3	627	564	357				
0	0	1024	164	245	359	2	2	2	20	697	836	3	8	17	472	283	283	5	7	5	805	777	176				
0	0	2048	198	262	178	2	2	2	28	572	567	3	8	17	261	199	97	5	7	7	253	203	194				
0	0	4096	2856	0	0	0	0	0	1	426	345	3	8	17	308	230	268	5	8	19	389	400	230				
0	0	8192	165	182	180	0	0	0	0	374	328	3	8	19	304	258	85	5	8	20	390	378	5				
0	0	16384	1403	157	157	0	0	0	0	1739	1185	178	3	9	14	325	273	274	5	8	15	240	202	355			
0	0	32768	202	250	347	0	0	0	0	552	504	185	3	9	18	219	168	274	5	8	21	276	293	174			
0	0	65536	264	250	0	0	0	0	9	177	115	175	3	9	22	309	279	86	5	8	23	144	153	196			
0	0	131072	403	477	358	0	0	0	0	11	252	303	0	3	10	348	313	73	5	8	12	165	199	174			
0	0	262144	238	210	0	0	0	0	13	679	781	0	0	10	19	247	281	0	5	8	16	160	107				
0	0	524288	200	340	401	0	0	0	0	15	410	458	1	0	21	203	172	70	5	8	19	237	247	359			
0	0	1048576	98	177	100	0	0	0	0	19	130	157	0	0	11	6	426	472	268	5	8	20	390	378	5		
0	0	2097184	66	34	170	0	0	0	0	21	312	410	179	0	0	11	6	314	319	96	5	8	11	290	280	173	
0	0	4194320	120	113	152	0	0	0	0	23	27	164	235	354	0	0	11	14	246	265	95	5	8	19	277	304	2
0	0	8388640	175	179	353	0	0	0	0	31	119	169	1	0	11	18	132	122	58	5	8	20	865	767	358		
0	0	16777280	184	174	0	0	0	0	42	172	166	174	0	0	11	22	190	199	270	5	8	20	340	345	0		
0	0	33554560	202	230	176	0	0	0	0	5	331	290	5	0	11	22	166	166	236	5	8	20	725	665	358		
0	0	67109120	142	112	84	0	0	0	0	11	447	407	178	0	0	11	14	212	233	100	5	8	20	119	104	187	
0	0	134218240	475	443	0	0	0	0	11	11	447	407	178	0	0	11	14	161	157	82	5	8	20	503	499	3	
0	0	268436480	66	34	276	0	0	0	0	15	155	144	175	0	0	11	18	74	61	72	5	8	20	342	318	357	
0	0	536872960	300	806	90	0	0	0	0	19	373	448	0	0	11	22	226	249	95	5	8	20	417	417	178		
0	0	107375840	317	320	87	0	0	0	0	21	202	169	349	0	0	11	16	136	139	97	5	8	20	88	153	11	
0	0	214751040	531	602	270	0	0	0	0	23	187	182	0	0	10	14	146	110	268	5	8	20	162	182	3		
0	0	429502400	202	214	266	0	0	0	0	27	242	241	221	0	0	10	16	421	428	0	5	8	20	120	120	173	
0	0	859004800	343	375	90	0	0	0	0	4	1371	1324	359	0	0	10	16	1700	2079	0	5	8	20	327	313	100	
0	0	171809600	251	170	259	0	0	0	0	12	1056	1056	329	0	0	10	16	360	360	28	5	8	20	352	333	281	
0	0	343619200	73	37	254	0	0	0	0	14	842	812	129	0	0	10	16	250	250	28	5	8	20	311	283	268	
0	0	686616640	561	568	93	0	0	0	0	23	286	410	356	0	0	10	16	253	158	157	5	8	20	228	235	3	
0	0	133272480	674	580	90	0	0	0	0	23	229	210	300	0	0	10	16	326	170	170	5	8	20	236	236	358	
0	0	266564640	883	862	90	0	0	0	0	29	70	30	183	0	0	10	16	241	241	207	5	8	20	327	292	280	
0	0	533129440	779	829	270	0	0	0	0	5	1133	1066	1	0	10	16	246	246	358	5	8	20	342	263	83		
0	0	1066460480	460	558	90	0	0	0	0	5	423	423	0	0	10	16	702	697	0	5	8	20	134	130	273		
0	0	213294480	304	310	72	0	0	0	0	5	243	204	356	0	0	10	16	1053	1117	0	5	8	20	327	287	253	
0	0	426409480	472	409	90	0	0	0	0	11	503	500	181	0	0	10	16	809	700	359	5	8	20	145	88	120	
0	0	852448440	484	459	269	0	0	0	0	13	527	541	2	0	10	16	163	167	0	5	8	20	230	263	96		
0	0	170449332	439	433	269	0	0	0	0	15	205	179	1	0	10	16	228	291	357	5	8	20	78	156	252		
0	0	340243312	493	511	89	0	0	0	0	17	129	107	1	0	10	16	607	241	241	5	8	20	214	164	76		
0	0	680249248	201	208	89	0	0	0	0	19	174	174	2	0	10	16	742	700	268	5	8	20	114	105	55		
0	0	134525248	262	262	262	0	0	0	0	21	291	291	159	0	0	10	16	320	320	92	5	8	20	1186	1523	0	
0	0	269178840	178	228	96	0	0	0	0	5	170	197	182	0	0	10	16	436	441	93	5	8	20	234	232	355	
0	0	538151424	219	225	260	0	0	0	0	7	163	179	174	0	0	10	16	277	258	269	5	8	20	257	233	165	
0	0	107193720	314	372	94	0	0	0	0	11	243	227	177	0	0	10	16	455	459	94	5	8	20	160	147	137	
0	0	214236256	236	255	257	0	0	0	0	14	542	497	359	0	0	10	16	256	246	246	5	8	20	178	164	197	
0	0	428623624	231	230	100	0	0	0	0	14	358	346	359	0	0	10	16	261	281	93	5	8	20	204	197	197	
0	0	856237200	194	225	263</																						

temperature factors was carried out on an Elliott 803B computer. After several cycles of refinement the value of R decreased to 15.9% at which stage changes in atomic parameters became less than the standard deviations.

The final atomic parameters are listed in column (ii) of Table 2 and observed and calculated structure factors are compared in Table 3, the mean value of $|F_o|$ for hkl and khl reflexions being given ($R=14.6\%$). Of the possible reflexions which were too weak to be recorded $|F_c|$ was found in each case to be less than the lowest observable $|F_o|$ and these data have not been used in the calculation of R and are not included in Table 3. Calculated intensities of powder reflexions are given in Table 1.

Discussion of the structure

The ideal structure is illustrated in Fig. 1, which shows successive layers of atoms parallel to the (001) face of the unit cell. The main shifts from idealized positions occur with the cadmium ions; each such ion is displaced about 0.20 Å or 0.25 Å towards neighbouring Cd vacancies. Other components of displacement are much smaller, being 0.05 Å.

The arsenic ions are but slightly displaced, relatively speaking, particularly As(1) and As(2) which lie on the twofold axes of the space group. As(5) ions almost lie on the fourfold screw axes and show the largest shifts; during the refinement process the parameters of As(5) ions were less inclined to converge to definite

Table 4. Bond lengths

Standard deviations are given in brackets.

Cd(1)-As tetrahedron	Cd-As(2)	2.87 (2) Å	As(2)-As(4)	4.49 (2) Å
	Cd-As(3)	2.95 (2)	As(2)-As(3)	4.44 (2)
	Cd-As(4)	2.62 (2)	As(2)-As(5)	4.32 (3)
	Cd-As(5)	2.51 (3)	As(3)-As(4)	4.53 (3)
			As(3)-As(5)	4.46 (3)
			As(4)-As(5)	4.49 (3)
Cd(2)-As tetrahedron	Cd-As(2)	2.71 (1)	As(2)-As(3)	4.44 (2)
	Cd-As(3)	2.65 (2)	As(2)-As(4)	4.49 (2)
	Cd-As(4)	3.01 (2)	As(2)-As(5)	4.66 (3)
	Cd-As(5)	2.70 (3)	As(3)-As(4)	4.30 (3)
			As(3)-As(5)	4.50 (3)
			As(4)-As(5)	4.57 (3)
Cd(3)-As tetrahedron	Cd-As(1)	2.67 (1)	As(1)-As(3)	4.51 (2)
	Cd-As(3)	2.69 (2)	As(1)-As(4)	4.56 (2)
	Cd-As(4)	2.65 (2)	As(1)-As(5)	4.50 (3)
	Cd-As(5)	3.20 (3)	As(3)-As(4)	4.58 (3)
			As(3)-As(5)	4.47 (3)
			As(4)-As(5)	4.58 (3)
Cd(4)-As tetrahedron	Cd-As(2)	2.68 (2)	As(2)-As(3)	4.57 (2)
	Cd-As(3)	2.67 (2)	As(2)-As(4)	4.53 (2)
	Cd-As(4)	2.67 (2)	As(2)-As(5)	4.32 (3)
	Cd-As(5)	3.02 (3)	As(3)-As(4)	4.55 (3)
			As(3)-As(5)	4.39 (3)
			As(4)-As(5)	4.46 (3)
Cd(5)-As tetrahedron	Cd-As(1)	2.86 (2)	As(1)-As(3)	4.51 (2)
	Cd-As(3)	2.66 (2)	As(1)-As(4)	4.46 (2)
	Cd-As(4)	2.98 (2)	As(1)-As(5)	4.50 (3)
	Cd-As(5)	2.65 (3)	As(3)-As(4)	4.53 (3)
			As(3)-As(5)	4.69 (3)
			As(4)-As(5)	4.46 (3)
Cd(6)-As tetrahedron	Cd-As(1)	2.73 (1)	As(1)-As(3)	4.51 (2)
	Cd-As(3)	3.00 (2)	As(1)-As(4)	4.46 (2)
	Cd-As(4)	2.67 (2)	As(1)-As(5)	4.48 (3)
	Cd-As(5)	2.55 (3)	As(3)-As(4)	4.30 (3)
			As(3)-As(5)	4.39 (3)
			As(4)-As(5)	4.51 (3)
Vacant tetrahedra	As(1)-As(3)	4.51 (2)	As(2)-As(3)	4.57 (2)
	As(1)-As(4)	4.56 (2)	As(2)-As(4)	4.52 (2)
	As(1)-As(5)	4.48 (3)	As(2)-As(5)	4.66 (3)
	As(3)-As(4)	4.55 (3)	As(3)-As(4)	4.57 (3)
	As(3)-As(5)	4.50 (3)	As(3)-As(5)	4.70 (3)
	As(4)-As(5)	4.49 (3)	As(4)-As(5)	4.52 (3)

values than were any of the other atomic parameters.

The present structure and that of von Stackelberg and Paulus are related geometrically to the structure of fluorite; this is illustrated in Fig. 2. Consider one fluorite-type unit cell with As^{3-} replacing Ca^{2+} and Cd^{2+} replacing F^- , but with two cube-diagonally opposite fluorite sites vacant. The unit cell of Goodyear and Steigmann contains 16 such units in the orientation shown in Fig. 2(a), the tail and head of the arrow being situated at the upper and lower vacant sites, respectively, in one unit. Fig. 2(b) shows the orientation of such units in the structure of Stackelberg and Paulus, there being 4 units in one unit cell of structure.

$\text{Cd}-\text{As}$ and $\text{As}-\text{As}$ bond lengths in each type of As tetrahedron are given in Table 4. The variation in some of the $\text{As}-\text{As}$ distances shows that there is appreciable distortion from cubic close-packing of As ions.

We wish to thank Dr D. W. G. Ballentyne and Mr G. L. Bucknell for providing us with specimens and Dr P. J. Wheatley, of Monsanto Research S.A., for permitting us to use some of his computer programs.

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A Refinement of the Structure of Calcium Hexa-antipyrine Perchlorate and a Comparative Study of Some Metal Hexa-antipyrine Perchlorates

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(Received 26 June 1967 and in revised form 23 October 1967)

Calcium hexa-antipyrine perchlorate, $\text{Ca}(\text{C}_{11}\text{H}_{12}\text{ON}_2)_6(\text{ClO}_4)_2$, is isomorphous with the corresponding magnesium and lead compounds and crystallizes in the trigonal space group $P\bar{3}$ with one formula unit in an elementary cell of dimensions $a=14.33$ and $c=9.78 \text{ \AA}$. The structure was refined by structure-factor least-squares method to an R value of 0.118 for 1131 observed reflexions. The structural features of magnesium, calcium and lead hexa-antipyrine perchlorates are compared. The coordination of antipyrine oxygen atoms around the metal ion is octahedral in all three cases; this octahedron becomes elongated along the $\bar{3}$ axis as the size of the metal ion increases. The nature of the metal-oxygen bonds is discussed on the basis of the known electronegativity coefficients of the relevant atoms and the observed metal-oxygen distances. In the antipyrine molecule, both the phenyl and the pyrazolone rings are planar and are inclined with respect to each other by angles varying from 62 to 68°. An attempt has been made to explain the observed bond lengths in the pyrazolone ring in terms of the major canonical structures proposed.

Introduction

The present study forms part of a program of systematic X-ray investigation of some metal antipyrine compounds undertaken in this laboratory to study the nature of the metal-oxygen bonding in these and also to elucidate the geometry of the antipyrine ring system.

Independent analyses of the isostructural lead and magnesium hexa-antipyrine perchlorates have already been reported (Vijayan & Viswamitra, 1966, 1967). Divalent calcium has an ionic radius ($r=0.99 \text{ \AA}$) mid-way between those of Mg^{2+} ($r=0.65 \text{ \AA}$) and Pb^{2+} ($r=1.21 \text{ \AA}$) (Wells, 1962) and hence it was thought worth while to analyse the structure of the calcium compound to facilitate a study of the changes introduced in the molecular geometry and packing as a result of the change in the size of the central metal ion. In addition, it would

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