

average Mn–O bond length of  $1.61 \pm 0.01$  Å. The e.s.d. is here set at a somewhat higher value to allow for the effect of some possible systematic errors. This bond length is in close agreement with the result of  $1.629$  Å found in  $\text{KMnO}_4$  by Palenik after applying a correction of  $0.022$  Å for anisotropic motion. Such a correction cannot be made in the present study on account of the incomplete thermal motion data.

The structure as seen in the [001] projection seems to consist of stacks of  $\text{MnO}_4$  tetrahedra and rows of Ag atoms extending in the *c* direction. The difference in *z* coordinates of the Ag and Mn atoms closest in projection is almost exactly  $\frac{1}{2}$ , and leads to a regular zigzag Ag–Mn–Ag–Mn chain running in the *c* direction. This description probably has no significance since there are closer Ag–Mn (and Ag–O) approaches in other directions. Each Ag atom is in fact surrounded by seven  $\text{MnO}_4$  groups with Ag–Mn distances varying between  $3.51$  and  $3.81$  Å, and corresponding close Ag–O approaches between  $2.33$  and  $2.67$  Å. The shortest Ag–Ag and Mn–Mn distances are  $3.63$  and  $4.33$  Å respectively, while the closest approaches between oxygen atoms in different permanganate groups vary from  $2.81$  to  $3.06$  Å. The shortest Ag–O and non-bonded O–O distances are markedly shorter than the corresponding distances in  $\text{KMnO}_4$  ( $2.62$  and  $3.06$  Å respectively). The present structure is clearly more compact since the molecular volume of  $\text{AgMnO}_4$  is about 15% smaller than that of  $\text{KMnO}_4$ . The closer approaches found in  $\text{AgMnO}_4$  compared to  $\text{KMnO}_4$  can probably be attributed to the fact that  $\text{AgMnO}_4$  is not a truly ionic compound, but contains a measure of covalent bonding. Unfortunately, accurate data for similar permanganates are not available.

The investigation of the structural aspects of the thermal decomposition of silver permanganate, which led to this structure determination, was undertaken at the suggestion, and under the initial guidance of Dr F.H. Herbstein. In the present work, helpful discussions with Dr G. Gafner are gratefully acknowledged.

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## The Crystal Structure of $\text{Cd}_3\text{As}_2$

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A Weissenberg study of the structure of  $\text{Cd}_3\text{As}_2$  has shown that the unit cell is tetragonal ( $a = 12.67 \pm 0.01$  Å,  $c = 25.48 \pm 0.02$  Å,  $Z = 32$ , space group  $I4_1cd$ ). Arsenic ions are approximately cubic close-packed and Cd ions are tetrahedrally coordinated. The main difference between the present structure and that proposed by von Stackelberg & Paulus lies in the disposition of the vacant tetrahedral sites. Both structures are geometrically related to the fluorite structure.

### Introduction

The crystal structure of  $\text{Cd}_3\text{As}_2$  was determined by von Stackelberg & Paulus (1935) as tetragonal ( $a = 8.95$ ,  $c = 12.65$  Å, space group  $P4_2/nmc$ ) with As ions approximately in a cubic close-packed array. Cd ions

were tetrahedrally coordinated and each As ion was surrounded by Cd ions at six of the eight corners of a distorted cube, the two vacant sites being at diagonally opposite corners of a cube face.

In the same paper von Stackelberg & Paulus also reported the crystal structure of  $\text{Zn}_3\text{As}_2$  as being the

same as that of  $\text{Cd}_3\text{As}_2$ , but with somewhat smaller cell parameters. However, a single-crystal study of  $\text{Zn}_3\text{As}_2$  by Cole, Chambers & Dunn (1956), using rotation and precession techniques, revealed a unit cell which was body-centred tetragonal, the probable space group being  $I4_1/acd$  with the  $a$  and  $c$  parameters, respectively,  $\sqrt{2}$  and 2 times those found by von Stackelberg & Paulus.

Two specimens of  $\text{Cd}_3\text{As}_2$  have been examined in the present work. The first was supplied by Dr D. W. G. Ballentyne, who suggested that the authors might care to re-examine the crystal structure of this material. The second was prepared by Mr G. L. Bucknell of this department as follows: Stoichiometric quantities of cadmium and arsenic were heated in a sealed evacuated silica tube at  $675^\circ\text{C}$  for 18 hours. The sample was then heated at  $850^\circ\text{C}$  for 2 hours prior to air quenching. The resulting polycrystalline mass was found to contain several single crystals sufficiently large for study by the Weissenberg technique. Both

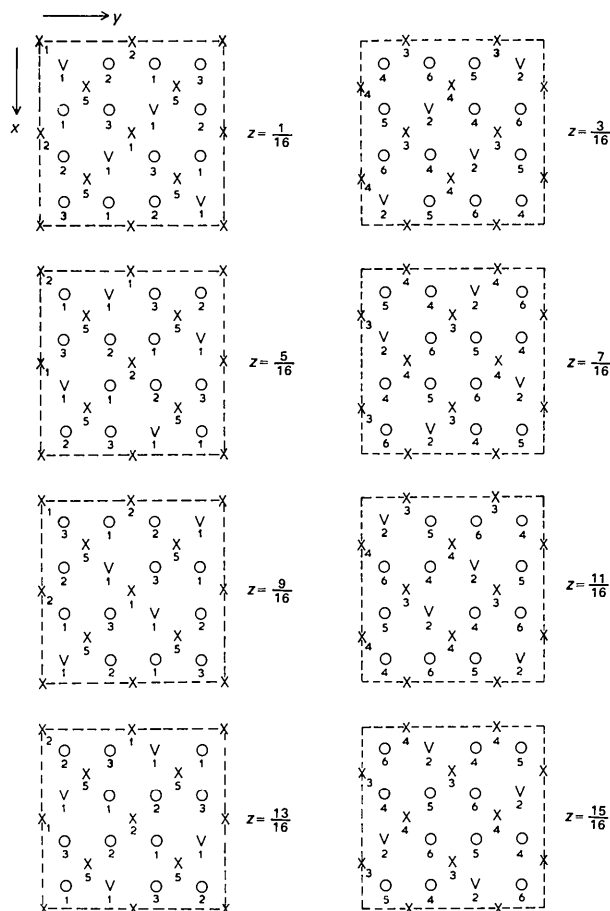


Fig. 1. Idealized structure of  $\text{Cd}_3\text{As}_2$ . Circles represent Cd atoms in the plane of the diagram. Crosses represent As atoms at  $\Delta z = \frac{1}{16}$  below the plane of the diagram. V1 and V2 are vacant Cd sites in first and second sets, respectively, of equivalent positions.

Table 1. Observed and calculated powder data

In the Table  $I_c \propto |F_c|^2 p f_1(\theta)$ , where  $p$  is the multiplicity and  $f_1(\theta)$  is the angular term defined by Goodyear & Duffin (1957). No correction has been made for absorption since this is nearly constant for all reflexions.

For the observed powder reflexions the reliability index

$$\Sigma |||I_o| - |I_c|| / \Sigma |I_o| = 0.090$$

$d_o$	$d_c$	$hkl$	$I_c$	$\Sigma I_c$	$I_o$	
7.35 Å	7.33 Å	112	49	49	57	
5.55	5.54	121	48	48	43	
4.72	4.71	123	127	127	94	
3.831	{ 3.837 3.822	{ 116 132	{ 137 225	362	465	
3.785	3.790	125	206			206
3.664	3.664	224	933	933	1114	
3.482	3.482	231	66	66	41	
3.248	3.247	233	317	317	297	
3.186	3.185	008	169	516	586	
3.167	3.167	040	347			
3.063	3.063	127	44			44
2.914	{ 2.915 2.908	{ 136 332	{ 278 83	361	316	
2.892	{ 2.893 2.889	{ 235 143	{ 7 486			493
2.527	2.528	237	70	70	40	
2.442	{ 2.443 2.439	{ 336 152	{ 131 17	148	130	
2.242	{ 2.246 2.240	{ 048 440	{ 1824 910			2734
2.141	{ 2.144 2.142	{ 12,11 156	{ 9 18	39	29	
1.932	{ 1.934 1.928	{ 356 23,11	{ 39 14			500
(vb)	1.912	{ 1.919 1.911	{ 22,12 264	{ 142 272		
	1.853	1.852	12,13	49		
1.842	{ 1.845 1.844	{ 455 363	{ 4 47	161	130	
1.831	{ 1.832 1.784	{ 448 11,14	{ 61 46			
(vb)	1.776	{ 1.779 1.774	{ 15,10 552	{ 24 11	161	115
	(vb)	1.707	{ 1.712 1.705	{ 23,13 273		
1.656		{ 1.657 1.651	{ 13,14 176	{ 83 68	283	306
1.647	1.647	275	131			
1.593	1.593	00,16	107	323	221	
1.583	1.584	080	216			

specimens were found to give identical X-ray powder patterns, and the powder data given in Table 1 are based on measurements of the patterns of both samples.

Since von Stackelberg and Paulus employed a rather different method of preparation (hydrogen and nitrogen gas transfer techniques) there is a small possibility that the disagreement between their structural data and those reported here is due to a difference in crystal structure rather than in quality of X-ray technique.

Weissenberg photographs of a suitable single crystal indicated a body-centred tetragonal cell approxi-

mately four times the size of the cell found by von Stackelberg & Paulus. An accurate determination of the lattice parameters, obtained from powder data, taken with Cu  $K\alpha$  radiation and a crystal-focusing camera of effective diameter 22.9 cm, gave  $a = 12.67 \pm 0.01$  and  $c = 25.48 \pm 0.02$  Å. This suggests the same relation to the data of von Stackelberg and Paulus as was found by Cole *et al.* for  $\text{Zn}_3\text{As}_2$ . Calculated and observed lattice spacings are compared in Table 1.

### Intensity measurements

The crystal selected for measuring intensities was of approximately square cross-section, of side 0.003 cm, and was elongated parallel to [110]. Sixteen layer lines were recorded on equi-inclination Weissenberg photographs taken about [110] with Cu  $K\alpha$  radiation.

Systematically absent reflexions were those for which  $h+k+l=2n+1$ ,  $2h+1=4n+1$  for  $hhl$ ,  $l=2n+1$  for  $0kl$  and possibly  $h=2n+1$  for  $hk0$ . These suggest  $I4_1cd$  or  $I4_1acd$  as the space group.

The cell and possible space groups permit more than 2000 reflexions within one octant of the sphere of reflexion for Cu  $K\alpha$  radiation. However, about 65% of these proved too weak to be recorded. Hence the intensities of 713 recorded reflexions were estimated visually from multiple film exposures using a calibrated scale. The observed intensities were corrected for Lorentz and polarization factors and were adjusted for spot shape on the upper layer lines (Phillips, 1954). Absorption corrections were made by assuming the crystal to be cylindrical, mean  $\mu r = 3.4$ , and applying the correction factors given by Bond (1959).

The intensities of the powder reflexions, given in Table 1 were obtained from a microdensitometer trace.

### The determination of the structure

From a consideration of the cell size and diameter, 4.44 Å (Clark, 1955), of the large As ion it follows that the As ions must be cubic close-packed, or very nearly so. For one face-centred cube of As ions, the cube side would be 6.3 Å; this is about one half of the  $a$  param-

eter (12.67 Å) and one quarter of the  $c$  parameter (25.48 Å) of the unit cell. It would appear then that the cell contains 16 such cubes, *i.e.* 64 As ions. With 32 formula units per cell the X-ray density is  $6.31 \text{ g.cm}^{-3}$ , which is in reasonable agreement with the observed macroscopic density of  $6.21 \text{ g.cm}^{-3}$  (*Handbook of Chemistry and Physics*, 1961). With the Cd ions in tetrahedral coordination, it is possible to arrange them around each As ion as in the structure of von Stackelberg & Paulus to give a structure consistent with the symmetry of the space group  $I4_1cd$ , the essential difference between the two structures being the orientation of the vacant cadmium sites. This distribution of Cd ions was not permitted in the other suggested space group,  $I4_1acd$ .

The ideal parameters of the proposed structure are given in column (i) of Table 2. Initial intensity calculations, using atomic scattering factors for the neutral atoms as listed in *International Tables for X-ray*

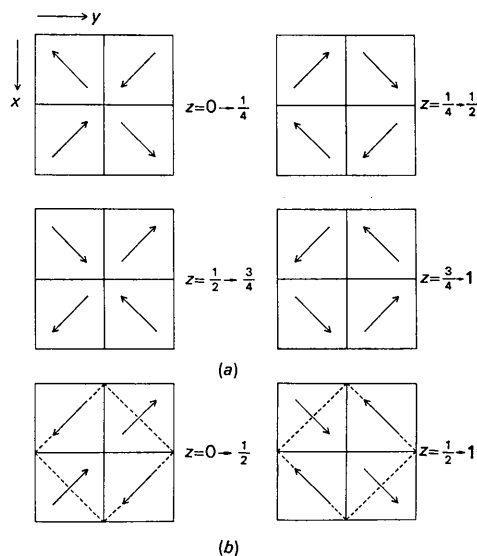


Fig. 2. Relation of the fluorite structure to (a) the present structure of  $\text{Cd}_3\text{As}_2$  and (b) the structure proposed by von Stackelberg & Paulus. In (b) the broken line indicates the base of the unit cell.

Table 2. Atomic parameters

(Origin on 2). Standard deviations are given in brackets.

Atom	Equipoint	(i) Initial			(ii) Final			
		x	y	z	x	y	z	B
Cd(1)	16(b)	$\frac{1}{2}$	$\frac{1}{2}$	$\frac{1}{6}$	0.3554 (11)	0.1208 (10)	0.0618 (05)	2.95 (21) Å <sup>2</sup>
Cd(2)	16(b)	$\frac{1}{2}$	$\frac{1}{2}$	$\frac{1}{6}$	0.1188 (09)	0.3584 (09)	0.0528 (04)	2.54 (19)
Cd(3)	16(b)	$\frac{1}{2}$	$\frac{1}{2}$	$\frac{1}{6}$	0.3933 (10)	0.3933 (11)	0.0727 (04)	2.18 (14)
Cd(4)	16(b)	$\frac{1}{2}$	$\frac{1}{2}$	$\frac{1}{6}$	0.1088 (11)	0.1059 (11)	0.1772 (04)	2.35 (15)
Cd(5)	16(b)	$\frac{1}{2}$	$\frac{1}{2}$	$\frac{1}{6}$	0.3799 (10)	0.1458 (10)	0.1892 (05)	2.86 (20)
Cd(6)	16(b)	$\frac{1}{2}$	$\frac{1}{2}$	$\frac{1}{6}$	0.1417 (09)	0.3797 (08)	0.1966 (04)	2.24 (17)
As(1)	8(a)	0	0	0	0	0	0.0001 (06)	1.42 (36)
As(2)	8(a)	0	0	$\frac{1}{2}$	0	0	0.2497 (06)	1.52 (37)
As(3)	16(b)	0	$\frac{1}{2}$	$\frac{1}{2}$	0.0072 (15)	0.2540 (11)	0.1231 (10)	2.18 (35)
As(4)	16(b)	$\frac{1}{2}$	0	$\frac{1}{2}$	0.2542 (10)	-0.0055 (13)	0.1259 (10)	1.85 (32)
As(5)	16(b)	$\frac{1}{2}$	$\frac{1}{2}$	0	0.2588 (10)	0.2399 (19)	-0.0005 (12)	1.41 (21)



temperature factors was carried out on an Elliott 803B computer. After several cycles of refinement the value of  $R$  decreased to 15.9% at which stage changes in atomic parameters became less than the standard deviations.

The final atomic parameters are listed in column (ii) of Table 2 and observed and calculated structure factors are compared in Table 3, the mean value of  $|F_o|$  for  $hkl$  and  $khl$  reflexions being given ( $R=14.6\%$ ). Of the possible reflexions which were too weak to be recorded  $|F_o|$  was found in each case to be less than the lowest observable  $|F_o|$  and these data have not been used in the calculation of  $R$  and are not included in Table 3. Calculated intensities of powder reflexions are given in Table 1.

### Discussion of the structure

The ideal structure is illustrated in Fig. 1, which shows successive layers of atoms parallel to the (001) face of the unit cell. The main shifts from idealized positions occur with the cadmium ions; each such ion is displaced about 0.20 Å or 0.25 Å towards neighbouring Cd vacancies. Other components of displacement are much smaller, being 0.05 Å.

The arsenic ions are but slightly displaced, relatively speaking, particularly As(1) and As(2) which lie on the twofold axes of the space group. As(5) ions almost lie on the fourfold screw axes and show the largest shifts; during the refinement process the parameters of As(5) ions were less inclined to converge to definite

Table 4. Bond lengths

Standard deviations are given in brackets.				
Cd(1)-As tetrahedron	Cd-As(2)	2.87 (2) Å	As(2)-As(4)	4.49 (2) Å
	Cd-As(3)	2.95 (2)	As(2)-As(3)	4.44 (2)
	Cd-As(4)	2.62 (2)	As(2)-As(5)	4.32 (3)
	Cd-As(5)	2.51 (3)	As(3)-As(4)	4.53 (3)
			As(3)-As(5)	4.46 (3)
		As(4)-As(5)	4.49 (3)	
Cd(2)-As tetrahedron	Cd-As(2)	2.71 (1)	As(2)-As(3)	4.44 (2)
	Cd-As(3)	2.65 (2)	As(2)-As(4)	4.49 (2)
	Cd-As(4)	3.01 (2)	As(2)-As(5)	4.66 (3)
	Cd-As(5)	2.70 (3)	As(3)-As(4)	4.30 (3)
			As(3)-As(5)	4.50 (3)
		As(4)-As(5)	4.57 (3)	
Cd(3)-As tetrahedron	Cd-As(1)	2.67 (1)	As(1)-As(3)	4.51 (2)
	Cd-As(3)	2.69 (2)	As(1)-As(4)	4.56 (2)
	Cd-As(4)	2.65 (2)	As(1)-As(5)	4.50 (3)
	Cd-As(5)	3.20 (3)	As(3)-As(4)	4.58 (3)
			As(3)-As(5)	4.47 (3)
		As(4)-As(5)	4.58 (3)	
Cd(4)-As tetrahedron	Cd-As(2)	2.68 (2)	As(2)-As(3)	4.57 (2)
	Cd-As(3)	2.67 (2)	As(2)-As(4)	4.53 (2)
	Cd-As(4)	2.67 (2)	As(2)-As(5)	4.32 (3)
	Cd-As(5)	3.02 (3)	As(3)-As(4)	4.55 (3)
			As(3)-As(5)	4.39 (3)
		As(4)-As(5)	4.46 (3)	
Cd(5)-As tetrahedron	Cd-As(1)	2.86 (2)	As(1)-As(3)	4.51 (2)
	Cd-As(3)	2.66 (2)	As(1)-As(4)	4.46 (2)
	Cd-As(4)	2.98 (2)	As(1)-As(5)	4.50 (3)
	Cd-As(5)	2.65 (3)	As(3)-As(4)	4.53 (3)
			As(3)-As(5)	4.69 (3)
		As(4)-As(5)	4.46 (3)	
Cd(6)-As tetrahedron	Cd-As(1)	2.73 (1)	As(1)-As(3)	4.51 (2)
	Cd-As(3)	3.00 (2)	As(1)-As(4)	4.46 (2)
	Cd-As(4)	2.67 (2)	As(1)-As(5)	4.48 (3)
	Cd-As(5)	2.55 (3)	As(3)-As(4)	4.30 (3)
			As(3)-As(5)	4.39 (3)
		As(4)-As(5)	4.51 (3)	
Vacant tetrahedra	As(1)-As(3)	4.51 (2)	As(2)-As(3)	4.57 (2)
	As(1)-As(4)	4.56 (2)	As(2)-As(4)	4.52 (2)
	As(1)-As(5)	4.48 (3)	As(2)-As(5)	4.66 (3)
	As(3)-As(4)	4.55 (3)	As(3)-As(4)	4.57 (3)
	As(3)-As(5)	4.50 (3)	As(3)-As(5)	4.70 (3)
	As(4)-As(5)	4.49 (3)	As(4)-As(5)	4.52 (3)

values than were any of the other atomic parameters.

The present structure and that of von Stackelberg and Paulus are related geometrically to the structure of fluorite; this is illustrated in Fig. 2. Consider one fluorite-type unit cell with  $\text{As}^{3-}$  replacing  $\text{Ca}^{2+}$  and  $\text{Cd}^{2+}$  replacing  $\text{F}^-$ , but with two cube-diagonally opposite fluorite sites vacant. The unit cell of Goodyear and Steigmann contains 16 such units in the orientation shown in Fig. 2(a), the tail and head of the arrow being situated at the upper and lower vacant sites, respectively, in one unit. Fig. 2(b) shows the orientation of such units in the structure of Stackelberg and Paulus, there being 4 units in one unit cell of structure.

Cd-As and As-As bond lengths in each type of As tetrahedron are given in Table 4. The variation in some of the As-As distances shows that there is appreciable distortion from cubic close-packing of As ions.

We wish to thank Dr D. W. G. Ballentyne and Mr G. L. Bucknell for providing us with specimens and Dr P. J. Wheatley, of Monsanto Research S.A., for permitting us to use some of his computer programs.

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## A Refinement of the Structure of Calcium Hexa-antipyrine Perchlorate and a Comparative Study of Some Metal Hexa-antipyrine Perchlorates

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Calcium hexa-antipyrine perchlorate,  $\text{Ca}(\text{C}_{11}\text{H}_{12}\text{ON}_2)_6(\text{ClO}_4)_2$ , is isomorphous with the corresponding magnesium and lead compounds and crystallizes in the trigonal space group  $P\bar{3}$  with one formula unit in an elementary cell of dimensions  $a = 14.33$  and  $c = 9.78$  Å. The structure was refined by structure-factor least-squares method to an  $R$  value of 0.118 for 1131 observed reflexions. The structural features of magnesium, calcium and lead hexa-antipyrine perchlorates are compared. The coordination of antipyrine oxygen atoms around the metal ion is octahedral in all three cases; this octahedron becomes elongated along the  $\bar{3}$  axis as the size of the metal ion increases. The nature of the metal-oxygen bonds is discussed on the basis of the known electronegativity coefficients of the relevant atoms and the observed metal-oxygen distances. In the antipyrine molecule, both the phenyl and the pyrazolone rings are planar and are inclined with respect to each other by angles varying from 62 to 68°. An attempt has been made to explain the observed bond lengths in the pyrazolone ring in terms of the major canonical structures proposed.

### Introduction

The present study forms part of a program of systematic X-ray investigation of some metal antipyrine compounds undertaken in this laboratory to study the nature of the metal-oxygen bonding in these and also to elucidate the geometry of the antipyrine ring system.

Independent analyses of the isostructural lead and magnesium hexa-antipyrine perchlorates have already been reported (Vijayan & Viswamitra, 1966, 1967). Divalent calcium has an ionic radius ( $r = 0.99$  Å) mid-way between those of  $\text{Mg}^{2+}$  ( $r = 0.65$  Å) and  $\text{Pb}^{2+}$  ( $r = 1.21$  Å) (Wells, 1962) and hence it was thought worth while to analyse the structure of the calcium compound to facilitate a study of the changes introduced in the molecular geometry and packing as a result of the change in the size of the central metal ion. In addition, it would

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